Formation of TiO₂ nanostructures modified Eumelanin films with enhanced properties for biopolymer implementations

Mushtaq Ali ^{a,b,*}, Muhammad Saad ^c, Mamatimin Abbas ^d, Arshad S. Bhatti ^e, Luca Ottaviano ^a,
Marianna Ambrico ^f, Andrea Di Cicco ^g, and Roberto Gunnella ^{g,h}

^a Dipartimento di Scienze Fisiche e Chimiche (DSFC), Università dell'Aquila, Via Vetoio 10 67100
L'Aquila, Italy

^b Department of Physics, HITEC University, Taxila Cantt, Pakistan

^c Institute of Physics, Kazan Federal University, Kazan, 420008 Russia

^d CNRS, Université Bordeaux, Bordeaux INP, Laboratoire de l'Intégration du Matériau au Système (IMS), UMR 5218, 16
Avenue Pey-Berland, 33607 Pessac Cedex, France

^e Department of Physics, COMSATS University Islamabad, 45550, Pakistan

^f CNR-Istituto di Nanotecnologia Via Amendola 122/D, I-70126 Bari, Italy

^g School of Science and Technology, University of Camerino, Via Madonna delle Carceri 62032 Camerino (MC),
Italy

^h INFN sez, Perugia Via A. Pascoli 06123 Perugia, Italy

* Corresponding author.

E. mail address: mushtaqali.zada@gmail.com (M.Ali)

28

29 **Abstract**

30 Thin films of hybrid melanin-TiO₂ nanoparticles (eumelanin: TiO₂) deposited from solution by
31 electro-spray were accurately inspected to unveil modified structural and electronic properties
32 for device implementations. Based on the reorganization of the melanin electronic valence
33 orbital, among other interesting behaviors, we observed up to a two orders increase in the
34 absorption coefficient in the visible range. Furthermore the red-shifted absorption features,
35 disclose that the extended π - stacking of oligomers is mainly responsible for the tunability of the
36 optical gap as confirmed by photoluminescence. TiO₂ nanostructures via oxidative
37 polymerization, further improve the oligomeric character observed by means of Raman
38 scattering. Finally, the photocatalytic activity of TiO₂ nanoparticles helps the saturation of
39 shallow trap states in melanin structure and consequently enhances the charge carrier transport.

40

41

42

43 **Keywords:** hybrid nanostructures, absorption features, spectroscopy, optoelectronics

44

45

46

47

48

49

50

51

52

53

54

1. Introduction

The recent advancements in the research of pigments like eumelanin (termed as melanin) are preluding to applications of these biopolymers in the fields of bioelectronics, sensors, and (opto)electronic devices [1-4]. Among the main physicochemical properties of melanin are the well-known strong UV-Vis optical absorption, relatively high ionic/electronic conductivity, photoconductivity (PC), exceptional biocompatibility, antioxidant and free-radical scavenging activities [3,5,6]. They are at the basis of the diverse functions of the natural melanin in the human body as a photoprotective agent and its involvement in some diseases like melanoma skin cancer, Parkinson's and Alzheimer neurodegenerative disease. However for study purpose and applications point of view, synthetic melanin guarantees a more convenient and reliable model instead of natural melanin due to inherent heterogeneity and difficulty in separation of the latter one [3,5]. Monomeric units of melanin structure are sketched in Fig. 1: (a) 5,6-dihydroxyindole (DHI or HQ), (b) 5,6-dihydroxyindole-2-carboxylic acid (DHICA), (c) indolequinone (IQ), and (d) semichinone (SQ), further details about melanin structure can be found elsewhere [3,6,7]. Previous studies started from characteristics of pellet type melanin samples [3], and subsequently developed in various directions, allowed the study of the secondary structure and the dependence of functional properties on the molecular organization and film quality [4,7,8]. Comprehensive results were obtained in terms of film smoothness, continuity and overall quality, followed by controlled structural, electronic and optoelectronic properties that have connected the fundamental features of the melanin organization, its feasible functionalizations [5,6,9] and practical applications [3,10-13].

An intriguing aspect of this research is involving the interactions of melanin with other organic or inorganic nanoparticles in hybrid/nanocomposite materials. Most common drawbacks are complete insolubility in organic solvents, the lack of appropriate deposition techniques that prevent the development of homogeneous and reproducible blended thin films, and the lack of a solid conceptual frame to understand the structure–property–function relationships upon the interaction of nanoparticles with synthetic melanin, hindering the full implementation of the bio(electronic) devices.

Titanium dioxide (TiO₂) nanoparticles are low cost, non-toxic, stable oxide semiconductor with distinctive physical and chemical properties, applied in some photovoltaic devices and several

85 photocatalysts applications. The combination of metal/metal oxide nanoparticles like TiO_2 with
86 organic structured materials is very attractive for the electronic, morphological properties
87 modifications [14-17] and towards practical applications. To date, the formation of hybrid
88 functional materials based on TiO_2 and monomer precursors, like DHICA were established using
89 different synthesis methods i.e., DHICA polymerization through coated TiO_2 nanoparticles and
90 solution mixed DHICA/ TiO_2 hybrid nanostructures [18,19], but complete picture of such hybrid
91 materials in the form of thin films, ideal to investigate nanohybrid materials is still lacking.
92 Hence, melanin- TiO_2 nanostructure is an open issue for understanding the mechanism of hybrid
93 formation, as their properties are strongly dependent on the particle size and aggregation
94 mechanisms that vary while passing from solution to the solid film.

95 Here, we report the fabrication of melanin blended TiO_2 nanoparticle thin films by solution
96 processed electrospray deposition (ESD) technique. The aim to develop thin films of these kind
97 of hybrid nanostructures was multifold: the foremost, is to enhance the photo absorption for
98 extended range of wavelength by exploiting the broadband absorption of melanin and TiO_2
99 nanoparticles. Also important is the response of melanin coupled to strong electron
100 acceptor/donor materials. Last but not least, the structure-function relation and mutual
101 dependence of these components must be critically investigated. By the way, properties of
102 melanin (particularly the transport) are complicated by the presence of two kinds of water, the
103 weakly and strongly bounded to the structure [3,6,9]. Adding the TiO_2 which has been widely
104 used for water splitting, purification or treatment applications will bring experimental
105 insights for the charge transport mechanisms and its response towards charge carrier
106 recombination activities. In addition, the motivation of this work was to explore in details the
107 characteristics of melanin and melanin: TiO_2 nanoparticle active layers, when heated at a high
108 temperature as required for many device preparations ($T > 120^\circ\text{C}$). To investigate all these issues,
109 we analyzed morphological, structural, electrical and optical properties of the active layer
110 melanin and blend (melanin: TiO_2) thin films for as prepared and treated samples, grown by ESD
111 deposition technique looking forward to organic/inorganic photovoltaics and other
112 (bio)electronic device applications.

Fig. 1. Monomeric units for melanin structure: (a) DHI or HQ, (b) DHICA, (c) IQ, (d) SQ. Red (oxygen atoms), light blue (nitrogen atoms), white (hydrogen atoms), and black (carbon atoms).

2. Experimental details

Synthetic melanin powder, prepared by oxidation of tyrosine with hydrogen peroxide (M8631 synthetic melanin) and TiO₂ nanoparticles (about 25 nm) were purchased from Sigma-Aldrich. Melanin is notoriously insoluble pigment, but dimethyl sulfoxide (DMSO) is the most effective solvent for its use. We dissolved melanin in Dimethyl Sulfoxide Methanol Solution (DMSO: CH₃OH; 1:20) with the concentration of 0.2 mg/ml. Similarly, for TiO₂ solution we used the same concentration of 0.2 mg/ml but different solvent (dichlorobenzene). After the preparation of the solution, the melanin and melanin: TiO₂ (with 1:1 ratio) were deposited by electrospray technique on indium-tin-oxide (ITO) and corning glass substrates. To realize the formation of hybrid nanostructures through enhanced interfaces area the melanin and TiO₂ proportion were kept the same, without going into the details of ratio optimization, which is essential for the purpose of efficient electronic devices in the future and is beyond the scope of this paper.

The schematic of ESD developed system [20,21] during thin films deposition is presented in the Fig. 2. A solution container was pressurized with the N₂ gas and the flow rate was controlled by measuring the time required to move a solution of 20 μL through 1 cm gap marked at various position on Teflon tube. The flow rate (1.3 μL/s) and applied voltage (2 kV) were kept constant during the deposition [6]. ESD melanin and blend films of the average thickness of 200 nm on

glass and ITO substrates were characterized with various microscopic and spectroscopic techniques.

Fig. 2. Schematic of the ESD system used for the film growth on respective substrates.

The morphological studies of these melanin and blend films were performed by Scanning Electron Microscopy (SEM- SU-1500 Hitachi) operating at 2–5 kV and Atomic Force Microscopy (AFM- Agilent's Pico Plus) working in tapping mode with cantilever frequency 160 kHz. The AFM images were analyzed quantitatively by Gwyddion Software [22]. Optical absorption spectra of melanin and blend films were measured in the ultraviolet, visible and near infrared regions. Absorption measurements were carried out by measuring the absorption light intensity obtained by spectrophotometer (Perkin, Elmer Lambda 950 UV/Vis/NIR). Photoluminescence (PL) study was carried out by using Dongwoo Optron W500i with He-Cd laser having wavelength 325 nm (3.8 eV) for excitation purpose.

The identification and investigation of functional group vibrations of melanin and blend (melanin: TiO₂) were performed using NICOLET 6700 Fourier Transform infrared spectroscopy (FTIR). Argon ion laser with laser lines, of wavelength 514 nm as excitation source was used in dark ambient at room temperature for Raman spectra (Renishaw 1000). Finally, the electrical characterization was taken under the dark and illumination conditions. The current–voltage (J – V) curves were obtained by using a Keithley 4200 semiconductor analyzer in dark and under illumination of an AM1.5 (100 mW/cm²) calibrated with IL1400BL radiometer. Gold stripes were evaporated with a channel length of 50µm onto glass substrates after film deposition to get

planar geometry for J – V measurements in both forward and backward voltage sweep directions.

3. Results and discussion

3.1. Film morphology

The morphology of synthetic melanin film is quite different and of better quality than that of natural melanin, due to the successive improvements in the methods of fabrication and the choice of solvents that affect the structure, and thin film quality [8,23,24]. In the present work, we report the thermal heat treatment effect on the morphology of synthetic melanin film and of its blend with TiO₂ nanoparticles. Previously [6], we found that the pure melanin thin films deposited at 3 kV applied voltage by ESD technique showed 2D regions with separated boundaries between flat zones formed on sample surface due to strain and to the process of drying. Therefore, here in this work we effectively lowered the applied voltage to 2 kV, for all the samples. SEM analysis show that the thin films fabricated at 2 kV by controlled electrospray deposition process [6,20] are quite homogenous, with complete flat coverage of the substrate with planar regular structures without any cracks or boundaries as shown in Fig. 3(a-d). SEM images for the annealed (at 120° C for 5 minutes) samples deposited on glass were presented in Fig. 3(a and b) for melanin and melanin: TiO₂ blended films respectively, while same annealed samples deposited on ITO substrate are in Fig. 3(c and d). Annealing is particularly important in the case of the hybrid films, to reach an acceptable homogeneity. The pristine hybrid film on ITO substrate (not shown here) indicated small clusters of TiO₂ nanoparticles coated and /or wrapped by melanin polymorphic organized layers. After annealing, the homogeneity of the blend film enhanced significantly and moderate distribution of TiO₂ nanoparticles (Fig. 3(b and d)) on the whole substrate was observed. Thermal annealing certainly affects the uniformity of thin films deposited on all kinds of substrates especially the conducting surfaces. In discussing samples deposited on glass substrates (upper panel) vs. samples on ITO substrates (lower panel) in Fig.3, the latter ones probably exhibited more wetting character in the blend film, preferentially due to the interaction between melanin and titania nanoparticles or interfaces with the substrate. These observations are favorable to conditions of molecular orbital hybridization and functionalization to be employed in real devices.

Fig. 3. SEM images comparison: (a) melanin film on glass substrate; (b) blend (melanin: TiO₂) on glass substrate; (c) melanin film on ITO substrate; (d) blend (melanin: TiO₂) on ITO substrate. Furthermore, all SEM images were from samples annealed at 120° C.

AFM images of melanin and blend (melanin: TiO₂) deposited on glass (upper panel) and ITO substrate (lower panel), after heat treatment are reported in Fig. 4, respectively. While comparing annealed melanin thin films grown on ITO with those on glass substrates, the first one shows clear improvement in the surface morphology, in terms of film roughness, homogeneity and oligomers structures planarity, as shown in Fig. 4(a and c). AFM images of blended titania films annealed at 120°C are reported in Fig. 4(b and d). Thermal annealing might affect polymers film thickness, by a more organized molecular spacing and in addition, helping supramolecular aggregation with increased uniformity and film quality. It is evidenced from the images that the signatures of titania nanoparticles are very clear, especially on ITO substrates, and showed almost no difference in root-mean-square roughness (before annealing 1.02 nm – after annealing 1.05 nm) associated with different regions of the blend film after annealing. In case of melanin films in the left panels of Fig. 4, the morphology disclosed overall uniform lengthened platelet [6,8]. Compared to the latter, slightly corrugated surface of melanin: TiO₂ samples (right panel of Fig. 4) was observed when blended with polymers by solution process, obviously due to

presence of titania particles agglomeration (although reduced with annealing) [15]. As a final observation, the films grown on conducting surfaces (Fig. 4d) with better adhesion are highly organized in terms of substructures, supported by the optical analysis in the coming section.

Fig. 4. AFM images in tapping mode of melanin thin film deposited on glass substrate (a), and melanin: TiO₂ blend on glass substrate (b) - (upper panel). AFM image of melanin thin film deposited on ITO substrate (c), and blend (melanin: TiO₂) deposited on ITO substrate (d) - (lower panel). Here, we presented the samples annealed at 120° C.

3.2. Optical absorption

The normalized absorption spectra of thin films melanin and its blend with TiO₂ as prepared on glass and ITO substrates were determined in the region (300- 950 nm) and shown in the Fig. 5(a and b), respectively. Before annealing the absorption spectra of melanin, particularly on glass substrates (Fig. 5(a), solid curve), showed the typical wide band character consisting of overall featureless smooth absorption without any apparent edge. However, an increase in absorption spectra in the visible wavelength range (380-750 nm) observed for pure melanin films, after annealing (Fig. 5(a and b), solid and dot curves) is suggesting more compacts films (especially on conducting substrate, probably due to adhesion properties of melanin [25,26]), and improved

self-assembly relating to monomeric components bonding. The most intriguing results were obtained after embedding TiO₂ nanoparticles in the melanin matrix particularly on conducting surfaces. The blended (melanin: TiO₂) dash and dash dot curves in Fig. 5(a and b) show enhanced absorption spectra compared to melanin, due to the photocatalytic activity of TiO₂ nanoparticles and better mutual adhesion during the ESD process [21,27]. Moreover, the normalized absorption spectra of melanin blending TiO₂ nanoparticles on ITO substrates showed (Fig. 5b) completely twisted behavior composed of new features different from the melanin characteristic broadband monotonic absorption [3,5]. These spectra with even broader and dominant absorption (350-800 nm) are similar to the prominent absorption of conventional low band gap conducting polymers in the range of 400-750 nm due to π - π bonding [15,20]. The peak observed near UV region at about 350 – 400 nm is due to the fundamental absorption of the Ti-O bond at that wavelength as shown in the inset of Fig. 5(a). Moreover, the wet nature of ESD thin films on ITO substrates reflect the slow growth rate for the dynamics of molecular organization/oligomerization, as compared to the fast growth rate on non-wet insulating glass substrates. These effects on conducting substrates were also accompanied by slight differences in morphology observed in SEM and AFM analysis. In addition, the photocatalytic properties of TiO₂ nanoparticles also played a significant role in the DHICA/DHI polymerization [19] that alters the absorption spectra of melanin in a different fashion from that of amorphous semiconductor [3]. Although melanin layer absorption features on conducting surface in the UV-Vis-NIR range are in agreement with the previous experimental observations [4] and theoretical simulations [28,29], and attributed to an increase in the degree of π -stacking and to the oxidation state of melanin, we emphasize here that the presently observed physical properties are the consequence of a controlled film fabrication by efficient ESD system able to blend melanin with TiO₂ nanoparticles which leads to a photocatalytic oxidative action on conducting surface. These examined absorption features are enhanced under thermal annealing with an increase of two orders of magnitude of absorption coefficient around 600 nm (Fig. 5b) in comparison to pure melanin suggesting the augmented adhesion between the two constituents. In fact, melanin similar to polydopamine [17,25] shows significantly strong adhesion to TiO₂ nanoparticles in the hybrid structure due to catechol groups.

We recently applied the Tauc model [4,24,30,31] for the calculation of energy gap value from the optical band edge as the absorption curve is not showing any of the exponential behavior of

Fig. 5. Normalized absorption spectra of melanin and its blend (melanin: TiO₂) films on glass and on ITO substrates are shown in Fig. (a) and (b), respectively. The absorption spectra of TiO₂ nanoparticles deposited on ITO substrate by ESD deposition is reported in the inset of (a) and energy band edges of melanin film on ITO is described in the inset of (b), respectively.

more disordered natural melanin. Equation of the form $(\alpha h\nu)^{1/2} \propto (h\nu - E_g)$ according to the Tauc model for indirect band gap semiconductor was used, where α is the absorption coefficient (obtained from the absorption curves in Fig. 5), $h\nu$ is the incident photon energy on the sample, and E_g gives the optical gap value by plotting the term $(\alpha h\nu)^{1/2}$ as a function of energy. The result of extrapolated onset on the horizontal axis of the absorption curves for a set of all films in this study was in the range of 1.25- 1.55 eV. The optical gap value 1.4 eV for melanin film deposited on ITO substrates was obtained using the Tauc plot as shown in the inset of Fig. 5(b). The estimated energy band gap values for melanin are slightly dependent on the choice of substrates, annealing and on the interaction of TiO₂ nanoparticles as photocatalytic that induced stacking ordering in the polymeric chains. These values are comparable to the melanin E_g values observed in the range from 1.0 to 1.4 eV [4,6,24,30,32,33,34]. We must be aware of the effect of Mie scattering that may be present at the longer wavelengths in the tail part of the spectra [3,4,6,9] affecting the energy band edge.

3.3. Photoluminescence spectroscopy

To examine further the optical properties of melanin and blend films of melanin: TiO₂ a PL study was carried out at room temperature in order to understand the interaction between the two components. As a comparison, Fig. 6(a and b) show the PL spectra of melanin and blend thin films excited with the excitation wavelength of 325 nm. It is found that different behavior of the PL spectra in the range of 300– 950 nm before annealing supports the significant substrate effects obtained previously from morphological and absorption analysis. The emission spectrum for 325 nm (3.8 eV) excitation reached the peaks maximum at around 550 nm obtained from melanin on glass substrates and then decreased monotonically with increasing emission wavelength [32,35], while the emission spectrum for melanin: TiO₂ blend on glass substrates (Fig. 6a) was reduced due to incorporation of the TiO₂ nanoparticles in the melanin matrix. Similar emission spectra related to the DHICA anions were obtained with maximum peak intensity at wavelength of 425 nm [19] compared to 550 nm in this case. Furthermore, it has been shown in the literature that melanin PL quantum yield is small, proved by the probe beam attenuation method and origin of some unexpected observations [7,26,36]. We supposed that the reason of unforeseen wide PL (FWHM ~ 1eV) behavior of pure melanin and blend films, on insulating substrates could be that the exciton emission originating from a hole-electron

Fig. 6. (a) PL spectra of melanin and blend samples deposited on glass substrates before annealing. (b) PL spectra of melanin and hybrid melanin: TiO₂ samples deposited on insulating and conducting substrates before and after annealing at 120° C for 5 minutes. All PL spectra were recorded with He-Cd laser having excitation wavelength 325 nm (3.8 eV).

recombination, was most probably due to trapped states on the surface [15], resulting from the combination of disorder of melanin monomers, defects (oxygen vacancies/structural defects due to agglomeration) and impurities including two kinds of water bonded in structure of melanin. Such impurities and defects could be eliminated by thermal, physical or chemical processes. In the case of thin films of organic species such as melanin and its blend with TiO₂ anatase phase over the ITO conductive (Fig. 6b) layer, electron and hole scavengers and the charge recombination mechanism in the visible and up to near infrared region are dominant [26,37,38]. The PL spectra after annealing (Fig. 6b) for a set of samples were shifted to an energy gap values near IR region (700- 900 nm) compared to that on glass substrates before annealing in the wavelength range 350-550 nm. Such observations are suggesting that the recombination of a hole with an electron due to surface trapped states/ disordering was reduced after annealing. The intense peaks in overall spectra for melanin and titanium dioxide were observed at the high wavelength near IR band gap region (Fig. 6b) for melanin, suggesting larger π -stacking oligomerization related to red-shifts in the absorption spectra and then emission. The intense excitation peaks in annealed PL emission spectra of the thin films started at offset of 1.3-1.45 eV and reach their maximum at 1.55 eV. These observations are in agreement with the estimated band gap values obtained from absorption analysis. The PL peaks (probably affected by spectrometer background) in the range of 1.5- 1.65 eV may correspond to different chromophores [3,7] of melanin DHI and DHICA, the latter one at lower energy compared to DHI that has dissimilar reaction times and extinction coefficients at different wavelengths [39]. In the Fig. 6(b), these observations further elucidate the trend of PL peaks intensity that quenched significantly in a systematic way from the pristine melanin/ITO film (curve-a) by annealing the same sample (curve-b) and then by adding TiO₂ nanoparticles as an electron acceptor with respect to the donor melanin material (curve-e). It reveals the strong interface formation between melanin and TiO₂ nanoparticles as a nanohybrid structures fostering robust devices applications. On the contrary, on insulating surface almost the same PL peaks intensity for thin films of melanin/glass (curve-c) and the blend/glass (curve-d) are observed, suggesting the weak interface formation between melanin and TiO₂ nanoparticles, that results in the TiO₂ nanoparticles PL response at around the wavelength values (450-550 nm).

To put into relation the absorption and PL spectra, we observed an increase in the absorbance with dominants absorption feature for melanin: TiO₂ samples and unusual emission spectra from

PL spectroscopy. The comparison of annealed melanin: TiO₂/ITO absorbance (in Fig. 5(b), dash-dot curve) and PL spectrum shown in Fig. 6 (curve-e) revealed our observations further. Here, we must point out that efforts are required to quench completely the PL spectra for the melanin-TiO₂ hybrid nanostructures peculiarly presented here in the near IR band gap region. For such a complex nanohybrid system, this could be done in the future by balancing the charge carrier mobilities while changing the ratio of melanin: TiO₂ (1:1) nanoparticles that will eventually reduce charge recombination processes.

3.4. Raman and Fourier transform infrared spectroscopy

Raman spectra were collected from melanin and its blend with TiO₂ (Melanin: TiO₂) films which were deposited on glass and ITO substrates through ESD technique. All ESD films investigated through visual inspection with microscope and by Raman spectroscopic analysis were quite homogenous, which permitted their study by a spatial resolution on the surface of the samples around 1 μm [6]. The Raman spectra of melanin are similar to the amorphous carbon graphitic structure having two dominant peaks at about 1600 cm^{-1} and 1300 cm^{-1} , which are related to the vibrational modes. These vibrational modes involving different atoms (carbon, oxygen, nitrogen and hydrogen) are organized according to different functional groups, bonded to indolic structure that gives the formation of basic monomeric units of melanin such as DHI, DHICA, SQ and IQ. The Raman spectra of thin films taken at room temperature consists of Raman active vibrational modes involving these different functional groups like, hydroxyl groups OH, carbonyl groups C=O, carboxylic acid COOH and NH groups, as reported in Fig. 7. In blended samples a slight shift was observed in main G ($\sim 1600 \text{ cm}^{-1}$) and D band ($\sim 1300 \text{ cm}^{-1}$) of carbon structures due to incorporation of TiO₂ nanoparticles of anatase phase. Moreover, the G and D band on ITO substrates were more intense than on glass. The Raman spectrum was fitted using the Gaussian function indicating the melanin (900-1800 cm^{-1}) and anatase TiO₂ (400-900 cm^{-1}) vibrational modes. The enclosed fitted area of blended Raman spectrum is reported in the left inset of Fig. 7(a) that indicates the presence of TiO₂ nanoparticles modes in the range of 400-900 cm^{-1} . The spectrum exhibit Raman fingerprints for anatase phase with important modes B_{1g} (419 cm^{-1}), A_{1g} (517 cm^{-1}) and E_g (639 cm^{-1}) in agreement with previous results [40].

In the recent literature, nine and eleven components fitted Raman spectrum has been already observed which were correlated with different vibrational modes of melanin [6,41]. In Fig. 7(a),

Fig. 7. (a) The normalized Raman spectra of blend (melanin: TiO₂) deposited on ITO and glass substrates, the inset showed the fitting plot of the experimental Raman spectrum along with chemical structure for both melanin (right) and TiO₂ (left), respectively. (b) Raman Spectra for melanin on ITO and glass substrates, before and after annealing of 120 °C.

Table 1. Raman Peaks related with modes in melanin blended TiO₂ nanoparticles complex system.

Raman Peaks (cm ⁻¹)	Assignments
Melanin Peaks in blend (melanin: TiO ₂)	
994 ± 2	O-H out of plane deformation
1078 ± 2	C-H deformation modes in or Out of plane
1218 ± 2	N-H in plane deformation and stretching C-C ring breathing
1290 ± 2	C-O stretching and OH deformation in COOH
1379 ± 2	Stretching (C-N)/indole ring vibration
1478 ± 2	C=C and C=N in plane vibration in pyrrole
1546 ± 2	Pyrrole ring stretching vibration/ C=C aromatic
1592 ± 2	Indole ring vibration / stretching antisymmetric (COO ⁻)
652 ± 2	C=C aromatic ring vibration
1737 ± 2	Stretching C=O in DHICA, Quinone or Ketone
TiO ₂ Peaks in Blend	
639 ± 2	E _g mode for anatase TiO ₂
517 ± 2	A _{1g} mode for TiO ₂
415 ± 2	B _{1g} mode for TiO ₂

the Raman spectra of blended films confirmed that titania nanoparticles are not perturbing the spectral response of melanin functional group. The best-fitting curve of melanin experimental Raman spectrum composed of ten components is shown in the right inset (zoomed from solid curve) of Fig. 7(a). The peaks around 800 to 1000 cm⁻¹ in the melanin structure are due to the deformation of modes of vibration in the pyrrole and indole structure. The broad peaks about 1400 cm⁻¹ overlap different vibrational modes namely C=C, C=N ring stretching in pyrrole structure. The lower intensity bands around 900 – 1200 cm⁻¹ are due to the vibration modes of CH, OH and NH with in plane or out of plane deformation. The broad band feature at 1550 and 1622 cm⁻¹ is the contribution to the Raman bands of pyrrole and indole ring stretching vibrations. The peaks centered at about 1700 and 1775 cm⁻¹ related to the C=O stretching vibrational modes in carboxylic acid or in quinone or ketone group. However, at 1775 cm⁻¹ and above weak peaks are observed corresponding to the stretching of carbonyl functional group vibrations in carboxylic acid [42]. The pyrrole di- and tri-carboxylic groups are also investigated during the subsequent oxidization of DHI and DHICA, which described the structure of melanin [43].

These Gaussian peaks value corresponding to the various vibrational modes for melanin and TiO_2 anatase peaks in the melanin: TiO_2 sample obtained from the fitting of the experimental Raman spectrum are listed in Table 1.

On the other hand, Raman spectra of the melanin on glass and ITO substrates before and after annealing are given in Fig. 7(b). It is important to note that the Raman spectrum of the melanin on glass reflect the structure like graphitic amorphous carbon, with a band D at 1300 cm^{-1} and a strong band G at 1600 cm^{-1} [6,41]. A small shift was observed in the G band position of melanin spectra deposited on ITO compared to glass substrates, and after annealing compared to as prepared melanin film on ITO. This regular small shift in the position of G strong band after annealing of melanin Raman spectra is similar to aromatic character changing from amorphous carbon graphitic structure to nanocrystalline diamond and/or nanocrystalline carbon graphitic structure. Furthermore, the shift in the G position is related to the optical Tauc gap [44], supporting our results obtained from absorption and PL analysis.

Fig. 8 depicts further the identification and investigation of functional group vibrations of melanin and melanin: TiO_2 hybrid nanostructures with a more comprehensive spatial range by using NICOLET 6700 Fourier Transform infrared spectroscopy (FTIR). The IR spectra indicated clearly that the large absorption due to functional group of melanin and hybrid melanin- TiO_2 nanostructures were characterized with different bonds in their respective regions as depicted on top of Fig. 8. The peak values assigned to different modes of vibrations at $1150\text{--}1371\text{ cm}^{-1}$ due to C-C, C-O, and at 1460 cm^{-1} due to C-N vibrations are similar to the reported values in references [45]. A wide band centered at $\sim 3000\text{ cm}^{-1}$ in both spectra was observed, which is corresponding to O-H stretching due to the presence of water in the melanin structure. IR absorption peaks in the range of $1460\text{--}1730\text{ cm}^{-1}$ are assigned to the bonding vibrations of $\text{C}=\text{C}$, $\text{C}=\text{O}$ and to the stretching of COOH double bonds. Moreover, in the blend thin film (melanin: TiO_2) the broad peaks observed at $3450\text{--}2400$, 1609 and 720 cm^{-1} were ascribed to the presence of OH, $\text{C}=\text{O}$ and Ti-O in the catalyst, and is suggesting the oxidative polymerization in the formation of hybrid nanostructure [46].

Fig. 8. IR absorption spectra of melanin thin films (lower-solid curve) and melanin: TiO₂ thin films (upper-dot curve) deposited by controlled ESD methods.

3.5. Electrical characterization

The charge transport characteristics of ESD synthetic melanin blended with TiO₂ nanoparticles films were measured under nitrogen- gas condition to avoid any surrounding effects. In Fig. 9 the Current–Voltage (J –V) curves were obtained by using a Keithley 4200 semiconductor analyzer in dark and under illumination of an AM1.5 (100 mW/cm²) calibrated with IL1400BL radiometer. Gold stripes were evaporated with a channel length of 50μm onto 7059 Corning glass substrates after film deposition to get planar geometry for J – V measurements in both forward and backward voltage sweep directions. In principle, more detailed analysis and improved performance could be obtained in the future by sandwiching the blended layer between two electrodes for an ideal devices modeling. At room temperature, the hysteretic behavior of melanin: TiO₂ films as shown in Fig. 9, contracted noticeably compared with hysteretic behavior that was reported in the previous work of ESD melanin films [6], due to the presence of shallow

trap states and to the existence of two types of water weakly and strongly bounded in melanin pores structures [9,12].

Previously, it was described that the electrical transport across the melanin active layers could strongly be affected by hydration state [3,5,9,12]. Such effects in the J-V curve for hybrid nanostructures layer were compensated by the presence of TiO_2 as an electron acceptor in the melanin matrix by filling the pores structures, while in case of ESD melanin films these trapped charges were released due to phonon-assisted hopping with increase in temperature. Moreover, the photocatalytic and water splitting characteristics of TiO_2 played crucial role to alter the hydration state for melanin charge transport mechanism in a reorganized structure. To assess the photo response of such active layer for organic/inorganic solar cells, the J-V curves for melanin: TiO_2 films under illumination were obtained. We found a strong photo response in the photoactive layer of melanin: TiO_2 thin film. The resistance of ESD melanin: TiO_2 film is very low at room temperature compared with ESD melanin films [9]. It is worth to note that TiO_2 nanoparticles as an electron acceptor semiconducting materials probably enhanced the charge transport towards electrodes by the reduced charge recombination effect in the active layer when

Fig. 9. J-V curve of a planar blend (melanin: TiO₂) structure both in forward and backward voltage sweep directions, measured by two Au electrodes separated by 50 μm. In the inset a planar geometry for J-V measurements on glass is shown.

added with melanin. These results are in agreement with optical, morphological and structural analysis, and revealed that hybrid melanin: TiO₂ nanostructures are active semiconducting agents that can be employed as potential biocompatible nanomaterials in organic/inorganic bio/(opto)electronic devices technology.

4 Conclusions

In this work, we described the formation of hybrid nanostructures of melanin with TiO₂ nanoparticles by ESD technique. Morphological analysis of the ESD thin films revealed quite homogeneous and uniform flat surfaces covering the complete substrate area. After subsequent thermal annealing, the oligomeric aggregates characteristic of melanin, as well as the hybrid nanostructures interfaces area are increased due to the mild distribution of titania particles in the melanin matrix and the strong adhesion between them that strongly influenced the physico-chemical properties. The broadband absorption of the melanin: TiO₂ hybrid films also increased as a result of the interaction of TiO₂ nanoparticles especially on the conducting substrates. In fact, after heat treatments the absorption features in the optical absorption and PL characteristics of melanin: TiO₂ layers on ITO substrates were enhanced. These facts preannounced the reorganized growth mechanism of self-assembled supramolecular structures with increasing degree of oligomerization and π -stacking of oligomers. In our opinion this result opens the way to a possible tuning of the band gap values and increase in the External Quantum Efficiency (EQE) in the infrared range of wavelengths. However, further quenching of the PL broad emission peaks is still an open issue for experimental researchers to balance the charge carrier mobility of optoelectronic devices with higher efficiency. These experimental results including: efficient charge transport mechanism, rationalized feature of absorption and quenching of PL emission, robust formation of melanin: TiO₂ hybrid nanostructures (elucidated by Raman and IR spectroscopies), represent a significant step toward future implementations of optoelectronic and bio-electronic/sensor devices using this important bio-polymer. .

Acknowledgments

Mushtaq Ali acknowledges the financial support of Higher Education Islamabad, under the Pakistan Program for Collaborative Research (PPCR-HEC) and the Department of Physics University of Camerino (MC), Italy. We are thankful to Assoc. Prof. N. Abbas for providing absorption spectroscopy facilities, Dr. Zia-ur-Rehman for FTIR and Dr. S. K. Shah for I-V technical assistance.

Author Information

*E-mail:mushtaqali.zada@gmail.com (M Ali)

REFERENCES

- [1] M. D’Ischia, A. Napolitano, A. Pezzella, P. Meredith, T. Sarna, Chemical and structural diversity in eumelanins: Unexplored bio-optoelectronic materials, *Angew. Chemie - Int. Ed.* 48 (2009) 3914–3921. <https://doi:10.1002/anie.200803786>.
- [2] P. Meredith, T. Sarna, The physical and chemical properties of eumelanin, *Pigment Cell Res.* 19 (2006) 572–594. <https://doi.org/10.1111/j.1600-0749.2006.00345.x>.
- [3] M. Ambrico, P.F. Ambrico, T. Ligonzo, A. Cardone, S.R. Cicco, M. D’Ischia, G.M. Farinola, From commercial tyrosine polymers to a tailored polydopamine platform: Concepts, issues and challenges en route to melanin-based bioelectronics, *J. Mater. Chem. C.* 3 (2015) 6413–6423. <https://doi:10.1039/c5tc00570a>.
- [4] C. Bonavolontà, C. De Lisio, M. D’Ischia, P. Maddalena, P. Manini, A. Pezzella, M. Valentino, Anomalous evolution of broadband optical absorption reveals dynamic solid state reorganization during eumelanin build-up in thin films, *Sci. Rep.* 7 (2017) 1–8. <https://doi:10.1038/s41598-017-00597-8>.
- [5] J. Wünsche, F. Cicoira, C.F.O. Graeff, C. Santato, Eumelanin thin films: Solution-processing, growth, and charge transport properties, *J. Mater. Chem. B.* 1 (2013) 3836–3842. <https://doi:10.1039/c3tb20630k>.

- [6] M. Abbas, M. Ali, S.K. Shah, F. D'Amico, P. Postorino, S. Mangialardo, M.C. Guidi, A. Cricenti, R. Gunnella, Control of structural, electronic, and optical properties of eumelanin films by electrospray deposition, *J. Phys. Chem. B.* 115 (2011) 11199–11207. <https://doi:10.1021/jp2033577>.
- [7] P. Meredith, B.J. Powell, J. Riesz, S.P. Nighswander-Rempel, M.R. Pederson, E.G. Moore, Towards structure-property-function relationships for eumelanin, *Soft Matter.* 2 (2006) 37–44. <https://doi:10.1039/b511922g>.
- [8] J.P. Bothma, J. De Boor, U. Divakar, P.E. Schwenn, P. Meredith, Device-quality electrically conducting melanin thin films, *Adv. Mater.* 20 (2008) 3539–3542. <https://doi:10.1002/adma.200703141>.
- [9] M. Abbas, F. D'Amico, L. Morresi, N. Pinto, M. Ficcadenti, R. Natali, L. Ottaviano, M. Passacantando, M. Cuccioloni, M. Angeletti, R. Gunnella, Structural, electrical, electronic and optical properties of melanin films, *Eur. Phys. J. E.* 28 (2009) 285–291. <https://doi:10.1140/epje/i2008-10437-9>.
- [10] M. Ambrico, A. Cardone, T. Ligonzo, V. Augelli, P.F. Ambrico, S. Cicco, G.M. Farinola, M. Filannino, G. Perna, V. Capozzi, Hysteresis-type current-voltage characteristics in Au/eumelanin/ITO/glass structure: Towards melanin based memory devices, *Org. Electron.* 11 (2010) 1809–1814. <https://doi:10.1016/j.orgel.2010.08.001>.
- [11] S.R. Cicco, M. Ambrico, P.F. Ambrico, M.M. Talamo, A. Cardone, T. Ligonzo, R. Di Mundo, C. Giannini, T. Sibillano, G.M. Farinola, P. Manini, A water-soluble eumelanin polymer with typical polyelectrolyte behaviour by triethyleneglycol N-functionalization, *J. Mater. Chem. C.* 3 (2015) 2810–2816. <https://doi:10.1039/c4tc01997k>.
- [12] M. Ambrico, P.F. Ambrico, T. Ligonzo, A. Cardone, S.R. Cicco, A. Lavizzera, V. Augelli, G.M. Farinola, Memory-like behavior as a feature of electrical signal transmission in melanin-like bio-polymers, *Appl. Phys. Lett.* 100 (2012) 253702. <https://doi:10.1063/1.4729754>.
- [13] M. Ambrico, P.F. Ambrico, A. Cardone, T. Ligonzo, S.R. Cicco, R. Di Mundo, V. Augelli, G.M. Farinola, Melanin layer on silicon: An attractive structure for a possible exploitation in bio-polymer based metal-insulator-silicon devices, *Adv. Mater.* 23 (2011) 3332–3336. <https://doi:10.1002/adma.201101358>.

- [14] A. Fakharuddin, F. Di Giacomo, A.L. Palma, F. Matteocci, I. Ahmed, S. Razza, A. D'Epifanio, S. Licoccia, J. Ismail, A. Di Carlo, T.M. Brown, Vertical TiO₂ Nanorods as a Medium for Stable and High-Efficiency Perovskite Solar Modules, *ACS Nano*. 9 (2015) 8420–8429. <https://doi:10.1021/acsnano.5b03265>.
- [15] J. Boucle, S. Chyla, M.S.P. Shaffer, J.R. Durrant, D.D.C. Bradley, J. Nelson, Hybrid solar cells from a blend of poly (3-hexylthiophene) and ligand-capped TiO₂ nanorods, *Adv. Funct. Mater.* 18 (2008) 622–633. <https://doi.org/10.1002/adfm.200700280>.
- [16] U.M. Garusinghe, V.S. Raghuwanshi, W. Batchelor, G. Garnier, Water Resistant Cellulose-Titanium Dioxide Composites for Photocatalysis, *Sci. Rep.* 8 (2018) 2306. <https://doi:10.1038/s41598-018-20569-w>.
- [17] H. J. Nam, B. Kim, M. J. Ko, M. Jin, J. M. Kim, and D.-Y. Jung, A New Mussel-Inspired Polydopamine Sensitizer for Dye-Sensitized Solar Cells: Controlled Synthesis and Charge Transfer, *Chem. Eur. J.* 2012, 18, 14000 – 14007. <https://doi: 10.1002/chem.201202283>
- [18] G. Vitiello, A. Pezzella, A. Zanfardino, B. Silvestri, P. Giudicianni, A. Costantini, M. Varcamonti, F. Branda, G. Luciani, Antimicrobial activity of eumelanin-based hybrids: The role of TiO₂ in modulating the structure and biological performance, *Mater. Sci. Eng. C*. 75 (2017) 454–462. <https://doi.org/10.1016/j.msec.2016.12.135>.
- [19] G. Vitiello, A. Pezzella, V. Calcagno, B. Silvestri, L. Raiola, G. D'Errico, A. Costantini, F. Branda, G. Luciani, 5,6-Dihydroxyindole-2-carboxylic Acid–TiO₂ Charge Transfer Complexes in the Radical Polymerization of Melanogenic Precursor(s), *J. Phys. Chem. C*. 120 (2016) 6262–6268. <https://doi:10.1021/acs.jpcc.6b00226>.
- [20] M. Ali, M. Abbas, S.K. Shah, E. Bontempi, P. Colombi, A. Di Cicco, R. Gunnella, Variability of physical characteristics of electro-sprayed poly (3-hexylthiophene) thin films, *Journal of Applied Physics*. 054515 (2011) 1–9. <https://doi:10.1063/1.3633519>.
- [21] M. Ali, M. Abbas, S.K. Shah, R. Tuerhong, A. Generosi, B. Paci, L. Hirsch, R. Gunnella, Realization of solution processed multi-layer bulk heterojunction organic solar cells by electro-spray deposition, *Org. Electronics*. 13 (2012) 2130–2137. <https://doi:10.1016/j.orgel.2012.06.016>.
- [22] D. Nečas, P. Klapetek, Gwyddion: an open-source software for SPM data analysis, *Open Physics*. 10 (2012) 181–188. <https://doi:10.2478/s11534-011-0096-2>.

- [23] G.S. Lorite, V.R. Coluci, M.I.N. Da Silva, S.N. Dezidrio, C.F.O. Graeff, D.S. Galvão, M.A. Cotta, Synthetic melanin films: Assembling mechanisms, scaling behavior, and structural properties, *J. Appl. Phys.* 99 (2006) 113511. <https://doi:10.1063/1.2200401>.
- [24] T. Ligonzo, M. Ambrico, V. Augelli, G. Perna, L. Schiavulli, M.A. Tamma, P.F. Biagi, A. Minafra, V. Capozzi, Electrical and optical properties of natural and synthetic melanin biopolymer, *J. Non. Cryst. Solids.* 355 (2009) 1221–1226. <https://doi.org/10.1016/j.jnoncrsol.2009.05.014>.
- [25] R. Mosurkal, Jin-An He, K. Yang, L. A. Samuelson, J. Kumar, Organic photosensitizers with catechol groups for dye-sensitized photovoltaics, *J. of Photochemistry and Photobiology A: Chemistry* 168 (2004) 191–196. <https://doi.org/10.1016/j.jphotochem.2004.05.004>
- [26] S.P. Nighswander-Rempel, J. Riesz, J. Gilmore, J.P. Bothma, P. Meredith, Quantitative Fluorescence Excitation Spectra of Synthetic Eumelanin, *J. Phys. Chem. B.* 109 (2005) 20629–20635. <https://doi:10.1021/jp053704+>.
- [27] M. Ali, M. Abas, S.K. Shah, E. Bontempi, A. Di Cicco, R. Gunnella, Film forming properties of electrosprayed organic heterojunctions, *Eur. Phys. Appl. Phys.* 62 (2013) 30202. <https://doi:10.1051/epjap/2013130107>.
- [28] D. Tuna, A. Udvarhelyi, A.L. Sobolewski, W. Domcke, T. Domratcheva, Onset of the electronic absorption spectra of isolated and π -stacked oligomers of 5, 6-dihydroxyindole: An ab initio study of the building blocks of eumelanin, *J. Phys. Chem. B.* 120 (2016) 3493–3502. <https://doi.org/10.1021/acs.jpcb.6b01793>
- [29] K.B. Stark, J.M. Gallas, G.W. Zajac, J.T. Golab, S. Gidanian, T. McIntire, P.J. Farmer, Effect of stacking and redox state on optical absorption spectra of melanins -- comparison of theoretical and experimental results., *J. Phys. Chem. B.* 109 (2005) 1970–1977. <https://doi:10.1021/jp046710z>.
- [30] L. Sangaletti, P. Borghetti, P. Ghosh, S. Pagliara, P. Vilmercati, C. Castellarin-Cudia, L. Floreano, A. Cossaro, A. Verdini, R. Gebauer, A. Goldoni, Polymerization effects and localized electronic states in condensed-phase eumelanin, *Phys. Rev. B.* 80 (2009) 174203. <https://doi:10.1103/PhysRevB.80.174203>.

- [31] I.G. Kim, H.J. Nam, H.J. Ahn, D.Y. Jung, Electrochemical growth of synthetic melanin thin films by constant potential methods, *Electrochim. Acta.* 56 (2011) 2954–2959. <https://doi:10.1016/j.electacta.2010.12.095>.
- [32] T. Eom, K. Woo, B.S. Shim, Melanin: A naturally existing multifunctional material, *Appl. Chem. Eng.* 27 (2016) 115–122. <https://doi:10.14478/ace.2016.1029>.
- [33] V. Capozzi, G. Perna, A. Gallone, P.F. Biagi, P. Carmone, A. Fratello, G. Guida, P. Zanna, R. Cicero, Raman and optical spectroscopy of eumelanin films, *J. Mol. Struct.* 744–747 (2005) 717–721. <https://doi:10.1016/j.molstruc.2004.11.074>.
- [34] L. Sangaletti, S. Pagliara, P. Vilmercati, C. Castellarin-Cudia, P. Borghetti, P. Galinetto, R. Gebauer, A. Goldoni, Electronic excitations in synthetic eumelanin aggregates probed by soft X-ray spectroscopies, *J. Phys. Chem. B.* 111 (2007) 5372–5376. <https://doi:10.1021/jp067415c>.
- [35] G. Perna, G. Palazzo, A. Mallardi, V. Capozzi, Fluorescence properties of natural eumelanin biopolymer, *J. Lumin.* 131 (2011) 1584–1588. <https://doi:10.1016/j.jlumin.2011.03.055>.
- [36] J.B. Nofsinger, T. Ye, J.D. Simon, Ultrafast Nonradiative Relaxation Dynamics of Eumelanin, *J. Phys. Chem. B.* 105 (2001) 2864–2866. <https://doi:10.1021/JP004045Y>.
- [37] J.V. Paulin, A.G. Veiga, Y. Garcia-Basabe, M.L.M. Rocco, C.F.O. Graeff, Structural and optical properties of soluble melanin analogues with enhanced photoluminescence quantum efficiency, *Polym. Int.* 67 (2018) 550–556. <https://doi:10.1002/pi.5543>.
- [38] S. Zhu, Y. Song, X. Zhao, J. Shao, J. Zhang, B. Yang, The photoluminescence mechanism in carbon dots (graphene quantum dots, carbon nanodots, and polymer dots): current state and future perspective, *Nano Res.* 8 (2015) 355–381. <https://doi:10.1007/s12274-014-0644-3>.
- [39] R. Micillo, L. Panzella, M. Iacomino, G. Prampolini, I. Cacelli, A. Ferretti, O. Crescenzi, K. Koike, A. Napolitano, M. D’Ischia, Eumelanin broadband absorption develops from aggregation-modulated chromophore interactions under structural and redox control, *Sci. Rep.* 7 (2017) 41532. <https://doi:10.1038/srep41532>.
- [40] N. Farahani, P. Kelly, G. West, M. Ratova, C. Hill, V. Vishnyakov, An Investigation into W or Nb or ZnFe₂O₄ Doped Titania Nanocomposites Deposited from Blended Powder

Targets for UV/Visible Photocatalysis, Coatings. 3 (2013) 153–165.
<https://doi.org/10.3390/coatings3030153>.

[41] G. Perna, M. Lasalvia, C. Gallo, G. Quartucci, V. Capozzi, Vibrational Characterization of Synthetic Eumelanin by Means of Raman and Surface Enhanced Raman Scattering, *Open Surf. Sci. J.* 5 (2013) 1–8. <https://doi.org/10.2174/1876531901305010001>.

[42] F. Rossella, P. Galinetto, M.C. Mozzati, L. Malavasi, Y. Diaz Fernandez, G. Drera, L. Sangaletti, TiO₂ thin films for spintronics application: a Raman study, *J. Raman Spectrosc. An Int. J. Orig. Work All Asp. Raman Spectrosc. Incl. High. Order Process. Also Brillouin Rayleigh Scatt.* 41 (2010) 558–565. <https://doi.org/10.1002/jrs.2465>.

[43] S.A. Centeno, J. Shamir, Surface enhanced Raman scattering (SERS) and FTIR characterization of the sepia melanin pigment used in works of art, *J. Mol. Struct.* 873 (2008) 149–159. <https://doi.org/10.1016/j.molstruc.2007.03.026>.

[44] A.C. Ferrari, J. Robertson, Interpretation of Raman spectra of disordered and amorphous carbon, *Phys. Rev. B.* 61 (2000) 14095. <https://doi.org/10.1103/PhysRevB.61.14095>.

[45] H. Okuda, A. Nakamura, K. Wakamatsu, S. Ito, T. Sota, Mid-infrared absorption spectrum of 5, 6-dihydroxyindole-2-carboxylic acid, *Chem. Phys. Lett.* 433 (2007) 355–359. <https://doi.org/10.1016/j.cplett.2006.11.058>.

[46] B. Erdem, R.A. Hunsicker, G.W. Simmons, E.D. Sudol, V.L. Dimonie, M.S. El-Aasser, XPS and FTIR Surface Characterization of TiO₂ Particles Used in Polymer Encapsulation, *Langmuir.* 17 (2001) 2664–2669. <https://doi.org/10.1021/la0015213>.

Mushtaq Ali: Conceptualization   Muhammad Saad: Data Curation Mamatimin Abbas: Editing ArshadS. Bhatti   Reviewing and Editing Luca Ottaviano   Reviewing and Editiong   Marianna Ambrico   Sample preparation   Andrea Di Cicco Revision, , and Roberto Gunnella: Experimental growth method, Reviewing and editing

Declaration of interests

X The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

X The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:

Camerino 19/07/2020

Roberto Gunnella

A handwritten signature in black ink, reading "Roberto Gunnella". The signature is written in a cursive style, with the first name "Roberto" and the last name "Gunnella" clearly distinguishable.